

# Electronic and Spectroscopic Study of the GaAs nitridation – Electronic Characterization Associated

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**Abstract:** Gallium nitride is one of the III-V semiconductors the most promising in many application areas. Indeed, because of its large direct bandgap (3,4 eV), it can be dedicated to both optoelectronic applications as transistors achieve hyper frequency. It allows the manufacture of components stable at high temperatures and high frequencies. In this work we are interested in the nitriding of gallium arsenide substrates of n-type in order to obtain a thin layer of GaN. The samples were chemically cleaned immediately introduced into the chamber ultra-high vacuum. They then undergo ionic cleaning in situ by argon ions. Spectrometric analyzes do show the absence of any impurities from the surface, in fact, it only detects the presence of gallium and arsenic. Before proceeding to the nitriding of our structures, deposition of gallium is conducted for 20 minutes to send them simultaneously nitrogen and gallium. Analyzes were performed elaborate structures using X-ray photoelectron spectroscopy (XPS). Deposition of mercury (Hg) has been developed on the structures, we have then made an electrical characterization I-V to see the effect of nitriding on the resulting structure.

**Key words:** GaAs, GaN, XPS, I-V characterization, Schottky diode.

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## 1. Introduction

A nitridation of GaAs is intensively studied in the last years as the first step of GaN layer deposition or a method for stabilization and passivation of a GaAs surface. Chemical and structural studies on a GaAs surface during and after nitridation.

In this paper, we are interested in the nitridation of GaAs to obtain a Schottky diode rectifier perfectly.

The I-V characteristic shows that our structure are Schottky diode rectifier whose electrical parameters are determined by a fitage of different regions of the characteristic; for our structure, we have determined the values of the series resistance, parallel resistance and the saturation current which are respectively  $13\Omega$ ,  $500\Omega$  et  $2,5 \cdot 10^{-6}$  A.

## 2. Sample Preparation

### 2.1 Cleaning samples :

The gallium arsenide substrates used were commercial substrates which are in the form of circular plates of thickness of  $400\ \mu\text{m}$  and a diameter of about 50 mm. These substrates were made by Czochralski pulling and then cut along the (100) orientation. These samples are doped n-type carrier concentration is  $N_d = 1,4 \times 10^{18}$  atoms/cm<sup>3</sup>. It is necessary to chemically clean the substrates as they contain various impurities such as oxides and fats.

The samples chemically cleaned were immediately introduced into the chamber ultra-high vacuum. They then undergo ionic cleaning in situ and characterized by photoelectron spectroscopy (XPS). All spectra were performed under excitation with X skate MgK $\alpha$  with energy  $h\nu = 1253,6$  eV.

To our experience, the ion cleaning with Ar<sup>+</sup> ions, is carried out with an argon ion kinetic energy of 1 keV, a current density of  $3,8\ \mu\text{A}/\text{cm}^2$  beam, argon pressure in the chamber of  $4 \cdot 10^{-5}$  Torr with a cleaning time of 60 min.

### 2.2 Study of the evolution of the Ga<sub>3d</sub> transition

Figures 1 and 2 show respectively the Ga<sub>3d</sub> transition after chemical cleaning and after cleaning ion respectively. We have broken this transition in a single peak as the difference in energy doublet Ga<sub>3d5/2</sub> Ga<sub>3d3/2</sub> related to the spin-orbit coupling is small (0,44 eV)<sup>[1, 2, 3]</sup>. The peak position associated with the GaAs link is located at  $19,2 \pm 0,2$  eV<sup>[4]</sup>. As shown in Figure 1, a GaAs substrate, although subjected to chemical cleaning has yet oxides. Indeed, there is a contribution located at  $20,4 \pm 0,2$  eV which corresponds to the Ga-O bonds attributed to gallium oxide Ga<sub>2</sub>O<sub>3</sub><sup>[4,5]</sup>. The analysis also shows that the transition ion bombardment of the substrate does not involve the formation of metallic gallium on the surface. In fact, the Ga<sub>3d</sub> transition has not a contribution of metallic gallium at  $0,7 \pm 0,2$  eV<sup>[4]</sup> towards lower bond energies. The parameters of the decomposition peak shape and width at half height of a peak in a given transition, are determined from the decay of the peak in the cleaned substrate. We considered that the width at half height of a transition is independent of the nature of the chemical bond. Note that the peak intensities of photoelectrons are measured relative to a reference sample or placed in the test chamber.

The decomposition parameters of the Ga<sub>3d</sub> transition are<sup>[1]</sup> :

- Width at half-height :  $1,8 \pm 0,2$  eV ;

- Peak shape : 20% gaussien – 80% lorentzien.

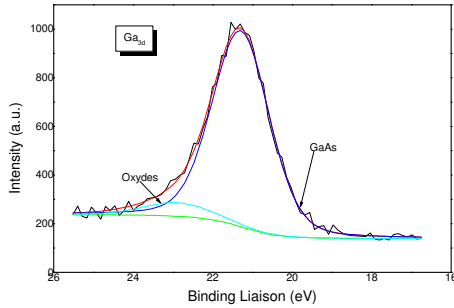


Figure 1: Spectrum of the Ga<sub>3d</sub> transition after chemical cleaning

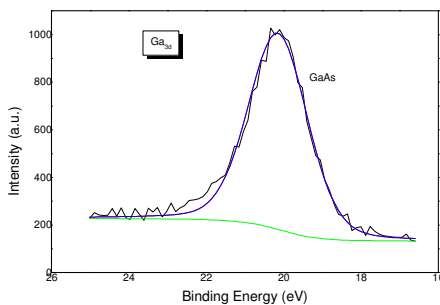


Figure 2: Spectrum of the Ga<sub>3d</sub> transition after ionic cleaning

### 2.3 Study of the As<sub>3d</sub> transition

This transition is shown in Figure 3. It represents the XPS spectrum of the As<sub>3d</sub> transition cleaning after 60 min with argon ions. It includes only the contribution As-Ga bonds which is located at  $40,8 \pm 0,2$  eV [4, 6]. This contribution is divided into doublet As<sub>3d5/2</sub> and As<sub>3d3/2</sub> related to spin-orbit coupling with the gap in the doublet is 0,7 eV and which the area ratio is between 1,2 and 1,4.

The parameters of the decomposition peak shape are [1] :

- Width at half-height :  $1,6 \pm 0,2$  eV ;
- Peak shape : 50% gaussien – 50% lorentzien.

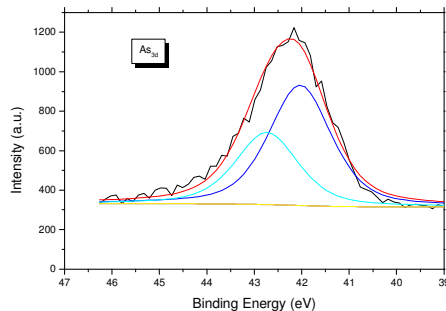


Figure 3: Spectrum of the AS<sub>3d</sub> transition after ionic cleaning

### 2.4 Study nitriding substrates

We studied the nitridation of GaAs using as a source of nitrogen production assets, the source type radio-frequency

plasma. We prepared samples nitrided at a temperature of 550 °C and a pressure of  $2 \times 10^{-4}$  Pa.

Figure 4 shows the spectrum of the Ga<sub>3d</sub> transition after nitriding. This transition has a new contribution at  $0,7 \pm 0,2$  eV towards higher binding energies. It corresponds to the Ga-N bonds [2, 7]. The parameters used for the decomposition Ga<sub>3d</sub> nitrided were kept identical to those of Ga<sub>3d</sub> cleaned.

This transition also includes a contribution due to oxides of gallium. The oxidation rate is 13% of the surface of the GaN bonds. This oxidation is attributed to experimental conditions such that the residual pressure of oxygen in the deposition chamber.

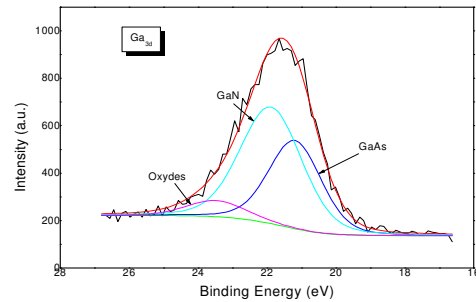


Figure 4: Spectrum of the Ga<sub>3d</sub> transition after nitridation

Figure 5 shows the N<sub>1s</sub> transition. It includes a contribution located at 397 eV associated with Ga-N bonds [8] and a contribution located at 2.5 eV to higher energies. This contribution is attributed to bonding states of (N-Ga-O) type content in the nitride film.

The decomposition parameters of the N<sub>1s</sub> transition are [1] :

- Width at half-height :  $2,2 \pm 0,2$  eV ;
- Peak shape : 50% gaussien – 50% lorentzien.

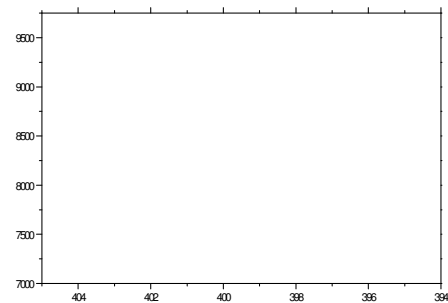


Figure 5: Spectrum of the N<sub>1s</sub> transition after nitridation

Figure 6 shows the As<sub>3d</sub> transition. It comprises in addition to the associated pair of links with the substrate, a contribution located at  $1,4 \pm 0,2$  eV, to higher energies. This contribution is probably due to connections arsenic oxides [9].

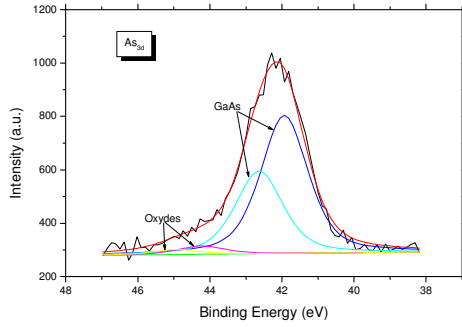


Figure 6: Spectrum of the  $As_{3d}$  transition after nitridation

### 3. Electrical characterization

#### 3.1 Numerical modeling :

The main process of electrical conduction observed in a Schottky diode are :

- The thermionic emission of carriers who can cross the barrier. The expression that connects the current potential applied directly on the diode is <sup>[10]</sup> :

$$I(V) = I(0) \times \exp\left(\frac{q \times V - R_s \times I}{n \times K_B \times T}\right) \quad (1)$$

Where  $V$  is the direct applied bias,  $n$  is the ideality factor,  $R_s$  is the series resistance of the diode,  $I(0)$  is the saturation current.

With ideality factor near 1.

- The recombination of carriers on deep centers generation-recombination. The current follows, again, the law of variation given by equation (1), with ideality factor near 2.
- The tunneling of electrons through the potential barrier of semiconductor to the metal. The tunnel effect can be observed, under direct polarization in the case of a degenerate semiconductor.

The equation of tunnel current is given by the following equation <sup>[10]</sup> :

$$I_{tu}(V) = I_{tu}(0) \times \exp\left(\frac{V - R_s \times I}{E_0}\right) \quad (2)$$

$$\text{where : } E_0 = E_{00} \times \coth\left(\frac{q \times E_{00} \times \pi}{k_B \times T}\right)$$

$$\text{With : } E_{00} = q \times \frac{\hbar}{2} \sqrt{\frac{N_d}{\epsilon_0 \epsilon_s m_e^*}}$$

- If there is a highly conductive layer at the metal-semiconductor interface, which is the origin of leakage currents in the diode, the current may be due to the effect of a parallel resistance  $R_p$ . The term of this current is <sup>[10]</sup> :

$$I_{R_p}(V) = \frac{V - R_s \times I}{R_p} \quad (3)$$

This is more likely when the level of reverse current is high enough.

### 3.2 Electrical measurements

To study the effect of nitriding of electronic devices based on GaAs, I-V characterization was made on nitrided GaAs chemically cleaned and non-nitrided. For an electronic device, a ball of mercury is applied onto the two structures. The result of the direct I-V characterization is shown in Figure 7. This figure shows that we have a current gain important for the structure based on GaAs nitrided structure which corresponds to a rectifier. We explain this in that the nitridation of GaAs has the effect of preventing oxidation of the surface. In addition, the GaN layer obtained is very thin ( $\approx 30 \text{ \AA}$ ), so the resulting structure is actually a Schottky diode Hg/GaAs with a GaAs surface treated by nitriding.

Figure 8 shows the direct and reverse I-V characteristic of the Hg/GaN/GaAs structure on a semi-logarithmic scale. This figure shows that the direct characteristic contains two parts corresponding to two different conduction models. For bias voltages greater than 0.55 V, the characteristic is dominated by the series resistance effect with a conduction due to the phenomenon of generation-recombination because the ideality factor is about 2. However, the reverse saturation current, the series resistor, the parallel resistor and the potential barrier height are respectively measured at :

$$I_s = 2,5 \cdot 10^{-6} \text{ A} ; R_s = 13 \ \Omega ; R_p = 500 \ \Omega ; \Phi_{BN} = 0,63 \text{ eV} .$$

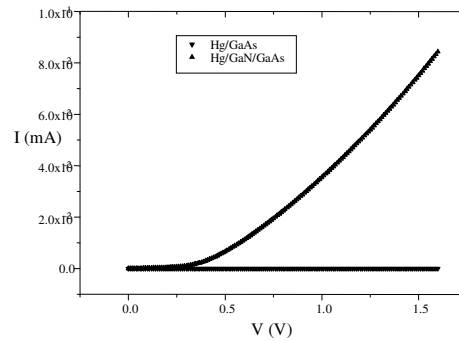


Figure 7: Direct I-V Characteristics for Hg/GaAs and Hg/GaN/GaAs structures

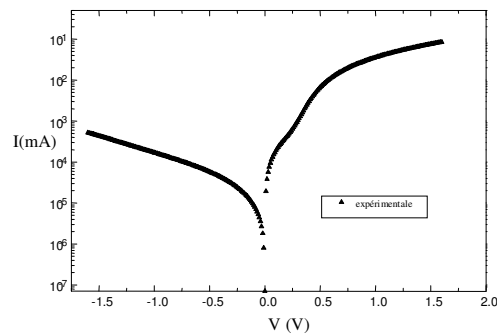


Figure 8: Direct and reverse I-V Characteristic for the Hg/GaN/GaAs structure

### 4. Conclusion

In this work, we have demonstrated the effectiveness of the process of nitriding of GaAs for obtaining the first GaN layers. We have demonstrated the interest of the photoelectron spectroscopy (XPS) for the control of surface evolution based treatments applied to the surface of the substrates. This

analysis also enables us to know the elements present on the surface as well as the elements which are connected. We noticed that the nitridation of GaAs at the same time causes the oxidation of the surface oxidation that we assigned to experimental conditions such that the residual pressure of oxygen in the deposition chamber.

I-V Electrical measurements on two structures type Hg/GaAs et Hg/GaN/GaAs ; the second structure looks rectifier with high current gain while the first structure did not. We also determine from the I-V characteristic obtained for the structure Hg / GaN / GaAs some electrical parameters of the developed structure.

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